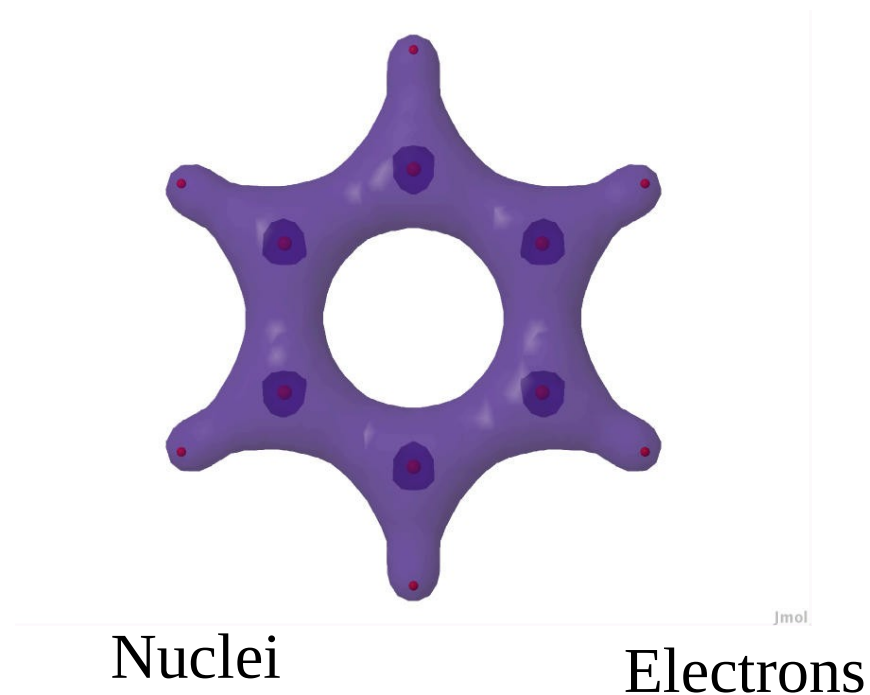
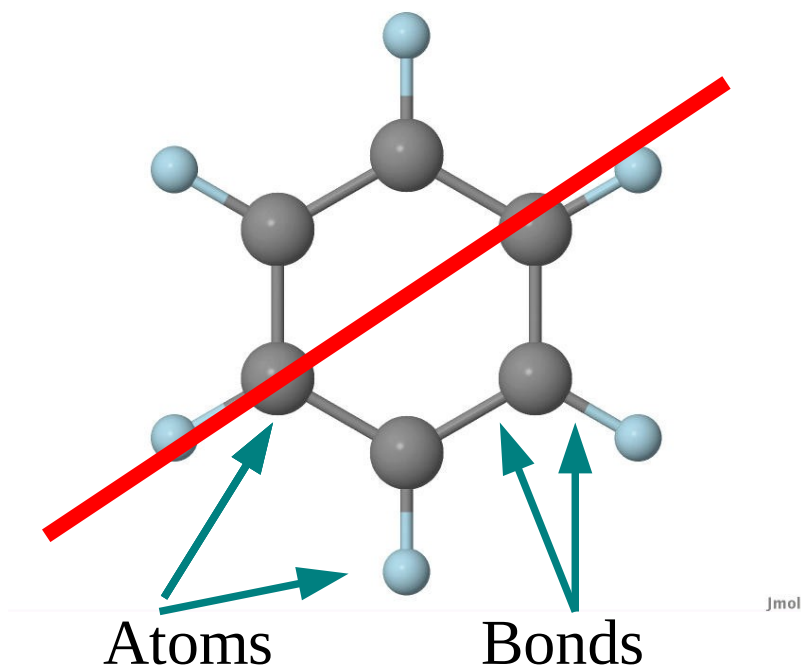


# First-Principles Vibrational spectroscopy and lattice dynamics of materials in the solid state

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Computational Science and Engineering  
Department  
*STFC Rutherford Appleton Laboratory*

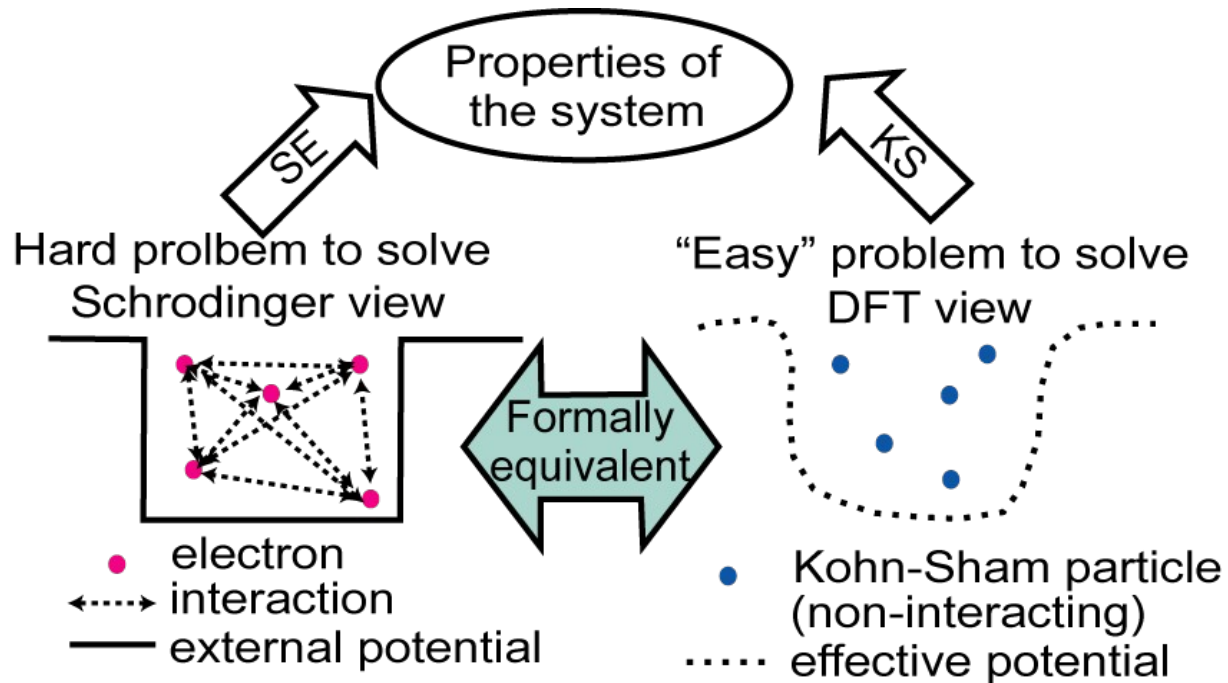


# First principles modelling



$$\frac{-\hbar^2}{2m_e} \nabla^2 \Psi + \hat{V} \Psi = E \Psi$$

# Density Functional Theory

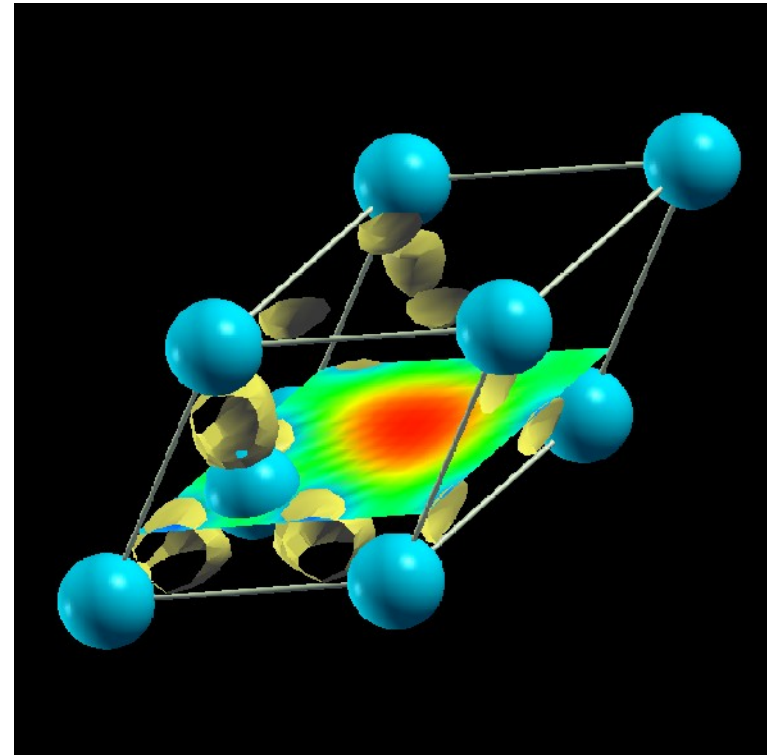


- Approximations to exchange and correlation
- Local Density Approximation (LDA)
  - Generalised Gradient Approximation (GGA)

Modified from Mattsson et al., (2005)  
*Modeling. Simul. Mater. Sci. Eng.* **13**, R1.

# DFT with a Plane-wave basis

- › Periodic boundary conditions applied to ions orbitals, electron densities.
- › Electron density represented on *grid*.
- › Basis coefficients of orbitals also stored on grid in  $\mathbf{G}$ -space.
- › Use FFTs to evaluate terms in Hamiltonian.
- › Never construct Hamiltonian, only compute effect of operator.
- › Kohn–Sham equations solved using SCF
- › PW basis is efficient, accurate, robust.
- › PW basis requires pseudopotentials.
- › Mature technology – good algorithms.
- › Robust in use, even by non experts.



# The CASTEP project

- Original 1990s CASTEP code by Mike Payne/Accelrys reached end of life.
- Complete re-engineering of a new plane-wave code from scratch beginning 1999.
- Core “Developer Group” of P. Hasnip, S. Clark, M. Probert, C. Pickard, M. Segall, P. Lindan, (Payne) and in 2002, K. Refson.
- Commercialised by Accelrys and integrated into Materials Studio
- Aim: build a flexible, well-engineered development platform for new physics using modular software practices and documented API specification.
- Parallel/HPC use built in from start. Release 1 in late 2001.
- Now at release 6.0.
- Excellent “Core” functionality with many additional capabilities; Structure, dynamics and many properties.

# CASTEP Capabilities

CASTEP has a variety of Hamiltonians and XC functionals

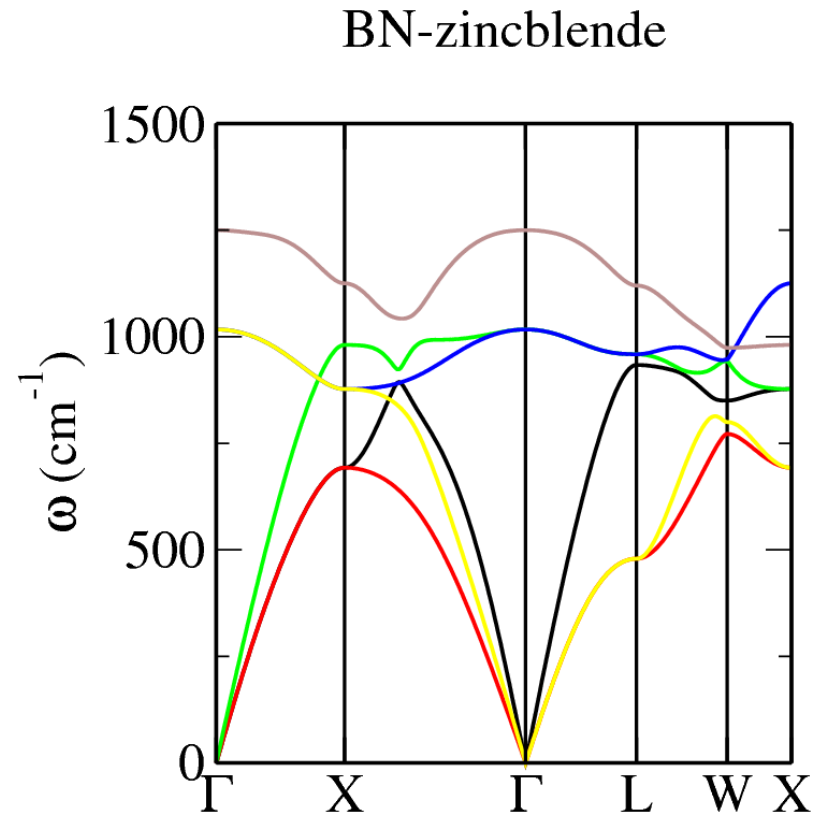
- *Pure local DFT (LDA, LSDA, PBE, RPBE, WC, ...)*
- *Hybrid HF exchange methods (HF, Screened HF, PBE0, B3LYP)*
- *Model methods (LDA+U)*
- *More under development (TDDFT, GW, ...).*

CASTEP can perform many spectroscopic calculations

- *IR and raman spectroscopy (vibrational/phonon)*
- *INS and IXS spectroscopy (vibrational phonon)*
- *Conduction-band optical dielectric spectra (EELS etc)*
- *Core level spectroscopy (ELNES, XANES)*
- *NMR chemical shifts*

# Vibrational Spectroscopy in CASTEP

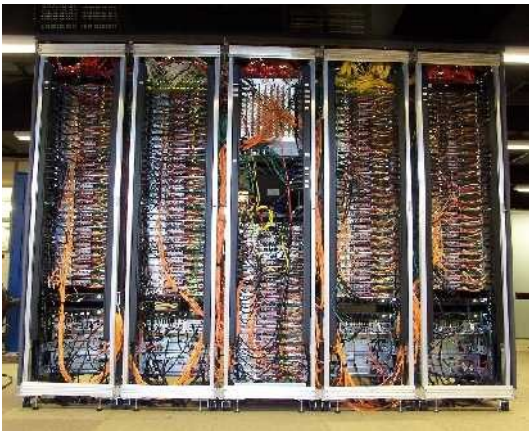
- Full *ab-initio* lattice dynamics code
- Plane-wave basis w pseudopotentials
- DFPT and supercell methods
- Phonons across full BZ by interpolation
- Symmetry analysis of eigenvectors
- Highly parallel for HPC use (can use 1000's of cores)





# Parallel Supercomputing

Parallel supercomputing enables large calculations and high throughput.



<http://www.hector.ac.uk>

<http://hpcsg.esc.rl.ac.uk/scarf/>

# Approaches to first principles Lattice Dynamics

## Density-Functional Perturbation Theory

- Works with LDA/GGA Hamiltonians
- Only primitive cell calculations required
- Fourier interpolation of dynamical matrices to entire BZ from grid of q-points.
- Can compute dielectric permittivity and Born effective charges
- LO/TO splitting included.
- IR absorptivity or Raman activity included

# Modelling the spectrum

## Orientationally averaged infrared absorptivity

$$I_m = \left| \sum_{\kappa, b} \frac{1}{\sqrt{(M_\kappa)}} Z_{\kappa, a, b}^* u_{m, \kappa, b} \right|^2$$

## Raman cross section

$$I_{\text{raman}}^m \propto |\mathbf{e}_i \cdot \mathbf{A}^m \cdot \mathbf{e}_s|^2 \frac{1}{\omega_m} \left( \frac{1}{\exp(\hbar \omega_m / k_B T) - 1} + 1 \right)$$

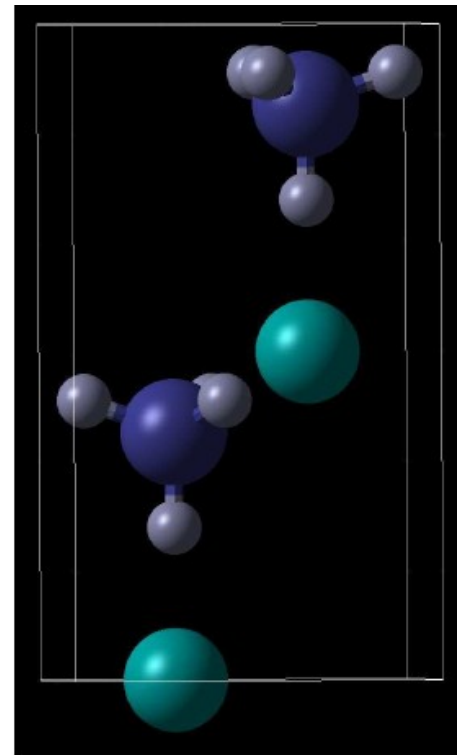
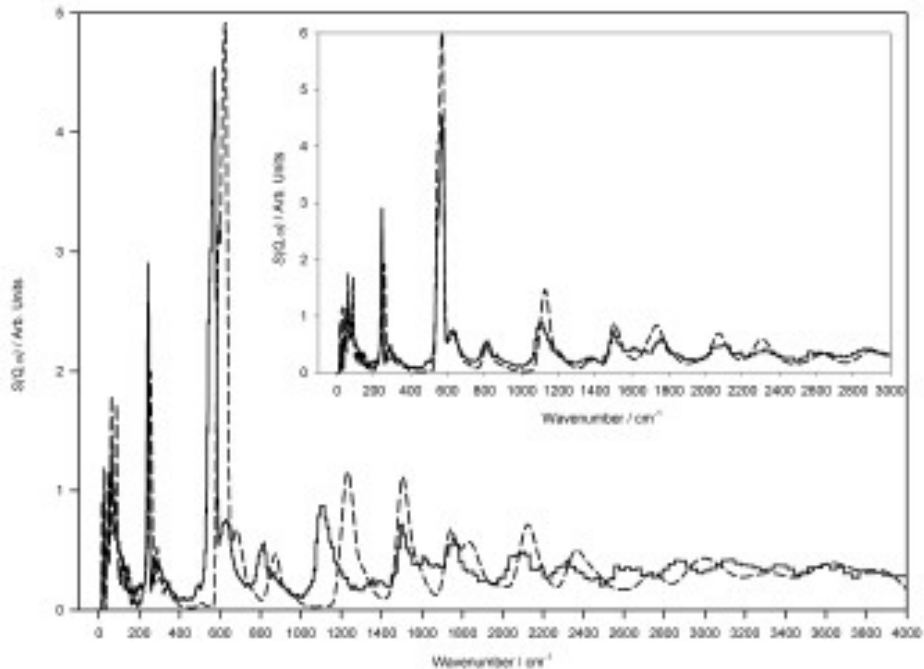
$$A_{\alpha, \beta}^m = \sum_{\kappa, \gamma} \frac{\partial^3 E}{\partial \boldsymbol{\epsilon}_\alpha \partial \boldsymbol{\epsilon}_\beta \partial u_{\kappa, \gamma}} u_{m, \kappa, \gamma} = \sum_{\kappa, \gamma} \frac{\partial \epsilon_{\alpha\beta}}{\partial u_{\kappa, \gamma}} u_{m, \kappa, \gamma}$$

## Inelastic neutron cross section

$$S^n(\omega_m) = \int d\mathbf{Q} \sum_{\kappa} \sigma_{\kappa} \left\langle \frac{(Q \cdot u_{m, \kappa})^{2n}}{n!} \exp(-(Q \cdot u_{m, \kappa})^2) \right\rangle$$

Spectral response to light depends on response of **electrons**; for neutrons only **nuclei**.

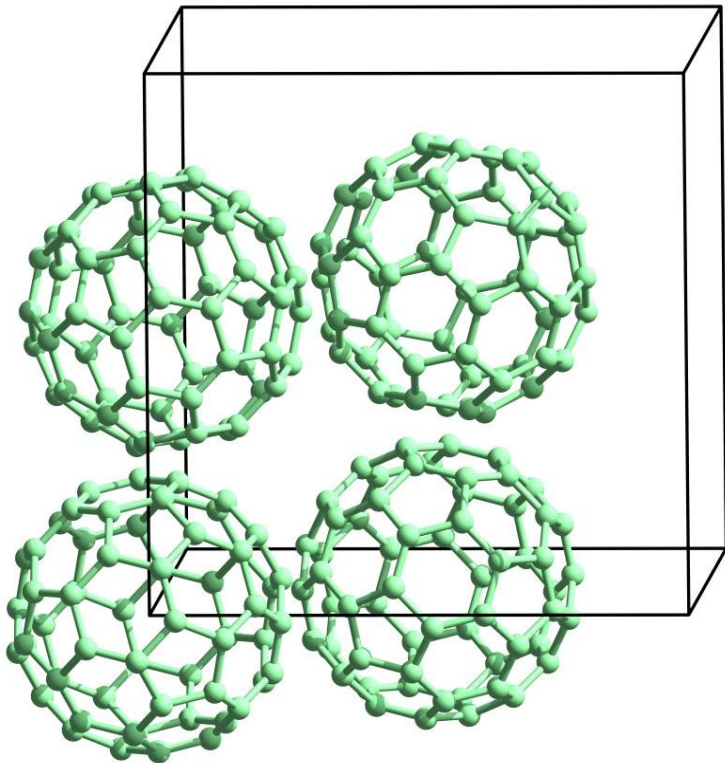
# INS Spectrum of Ammonium Fluoride



- $\text{NH}_4\text{F}$  studied in the ISIS TOSCA spectrometer.
- Collaborator: Mark Adams (ISIS)
- Structurally isomorphous with ice Ih
- INS spectrum modelled using ACLIMAX software (A. J. Ramirez Cuesta)
- Predicted INS spectrum agrees with experiment
- $\text{NH}_4$  libration modes in error by ? 5%.
- Complete mode assignment achieved.

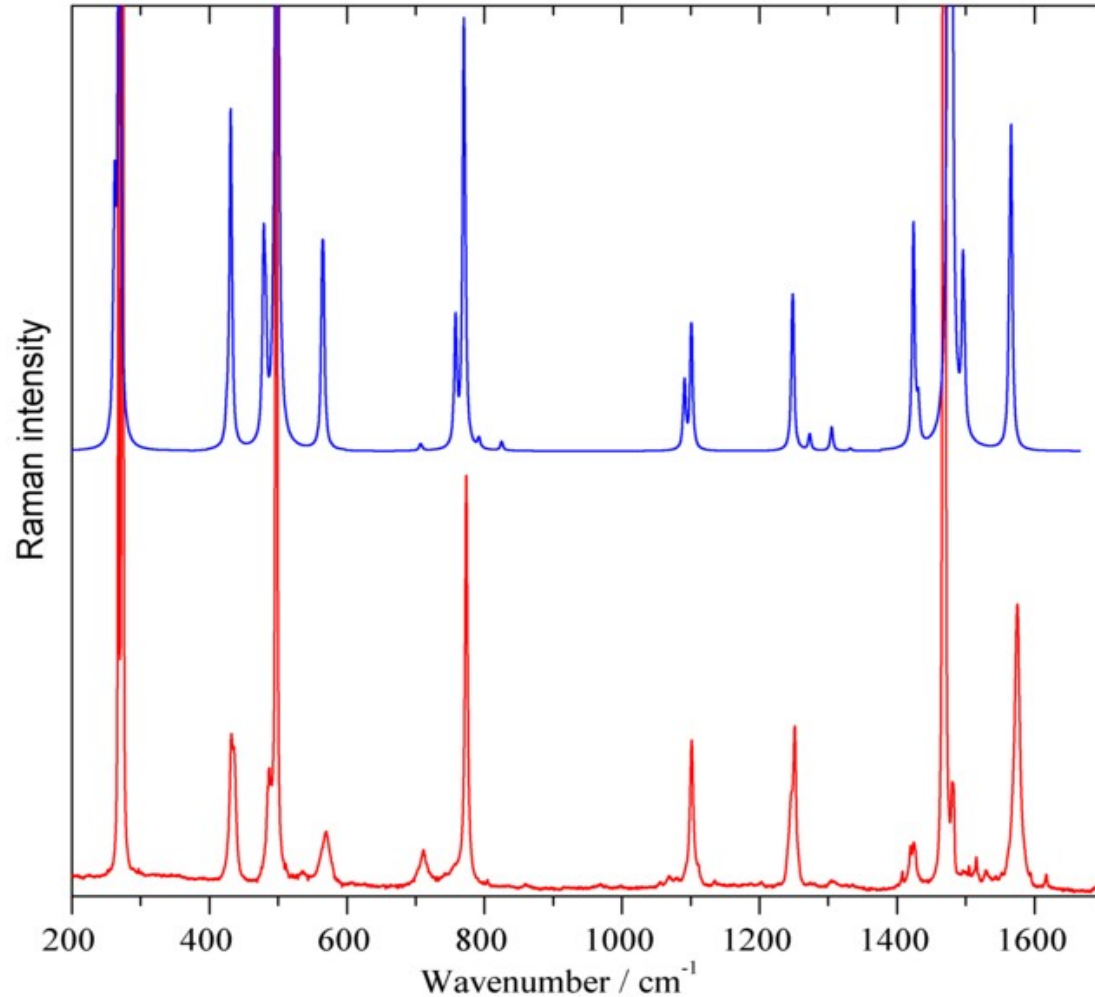
*Adams, Refson & Gabrys, Phys. Chem. Chem. Phys. 7, 3685 (2005)*

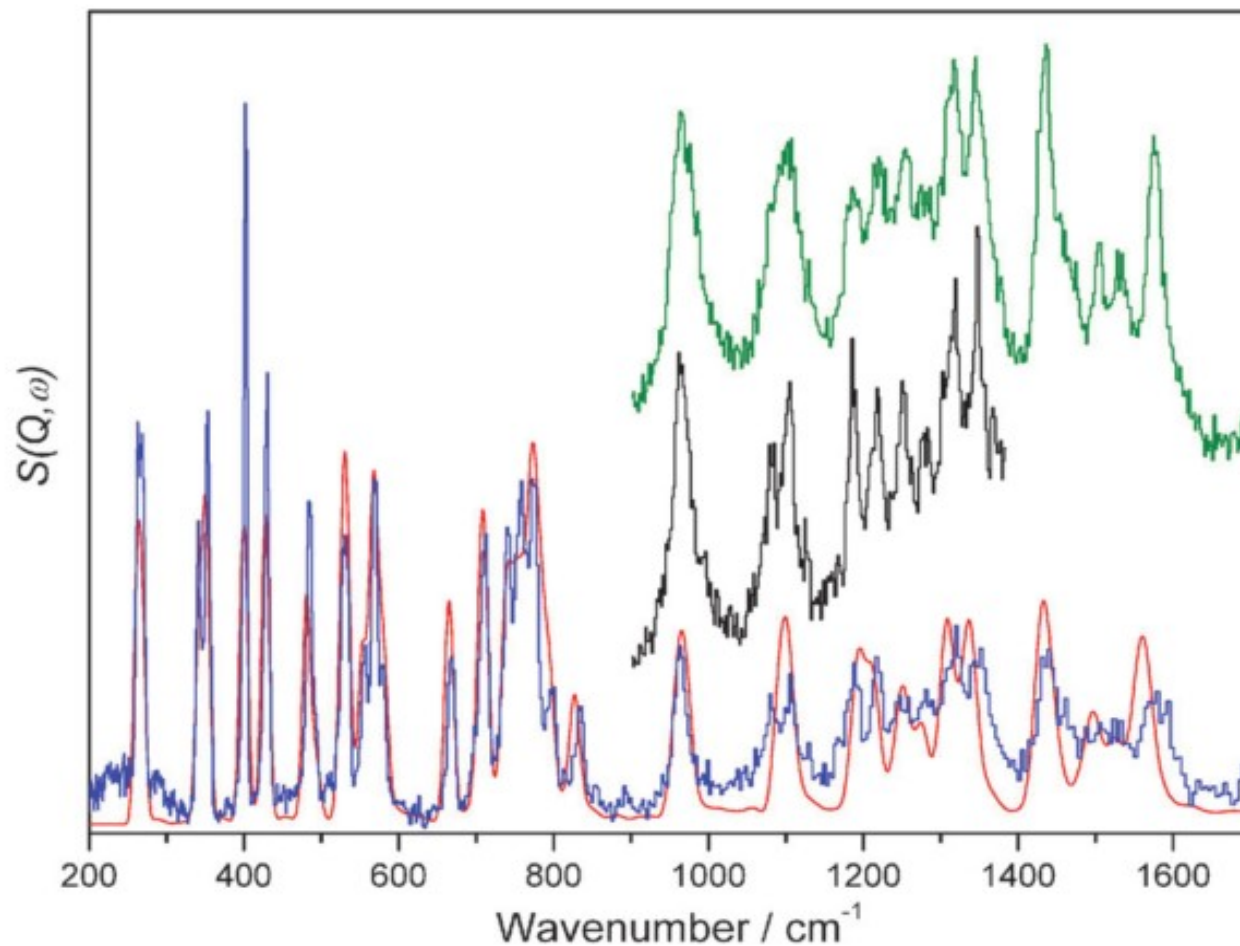
# Raman and ir spectroscopy of $C_{60}$



- Above 260K takes Fm3m structure with dynamic rotational disorder
- $m\bar{3}m$  point group lower than  $I_h$  molecular symmetry
- Selection rules very different from gas-phase.
- Intramolecular modes and factor group splitting seen.
- Try ordered Fm3 model for crystal spectral calculation.
- Full CASTEP DFPT lattice dynamics calculation on Fm-3 and Pa3 phases
- Complete assignment of vibrational modes [Phys Chem Chem Phys (2011).]

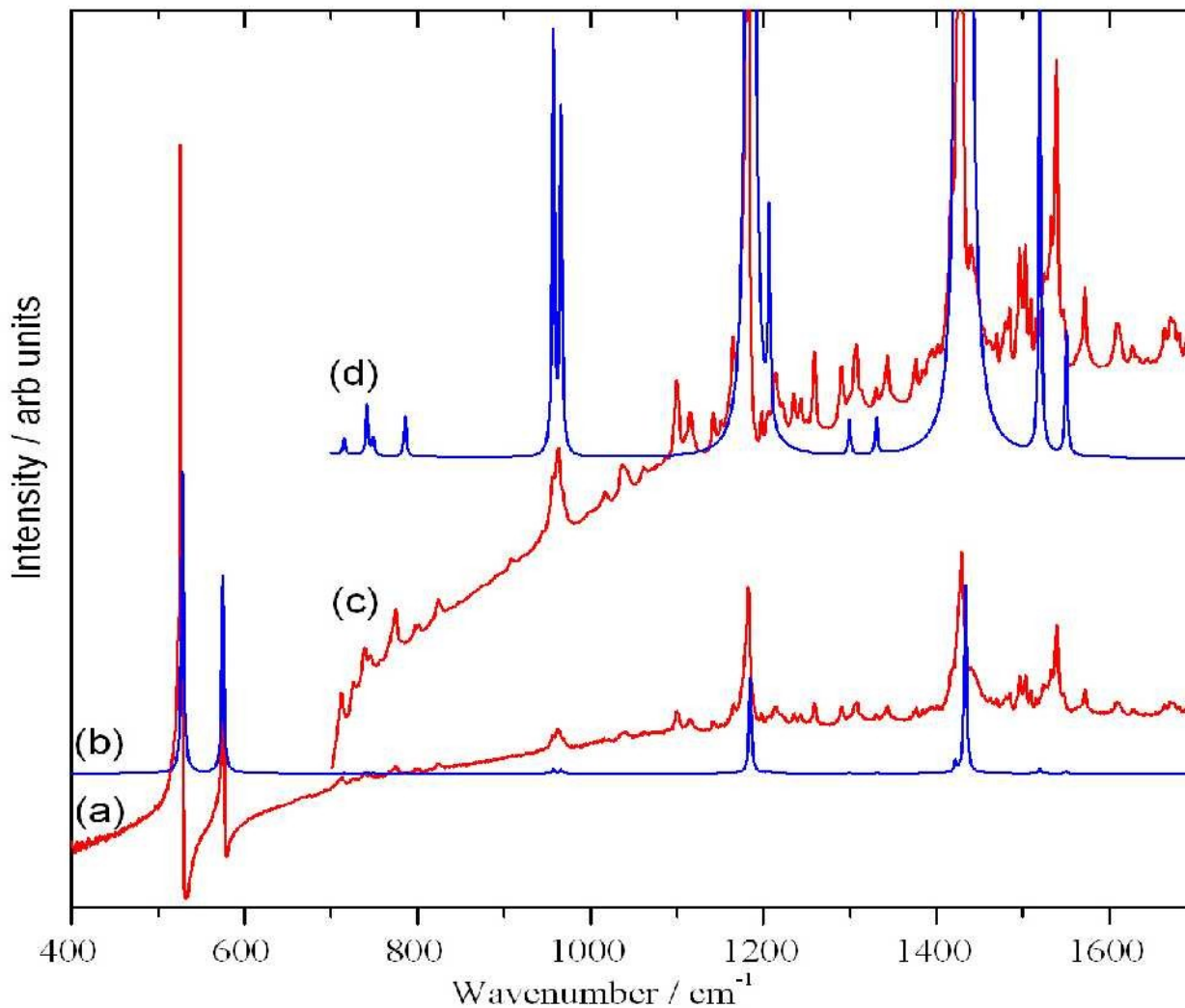
# GGA Raman spectrum of $C_{60}$





**Fig. 6** INS spectra of the internal modes of  $C_{60}$  in the  $Pa\bar{3}$  phase: recorded on TOSCA at 20 K (blue), recorded on MARI at 5 K with 1815  $\text{cm}^{-1}$  (olive green) and 1452  $\text{cm}^{-1}$  (black) incident energy compared with that generated from the CASTEP calculation (red).

# GGA infrared spectrum of $C_{60}$



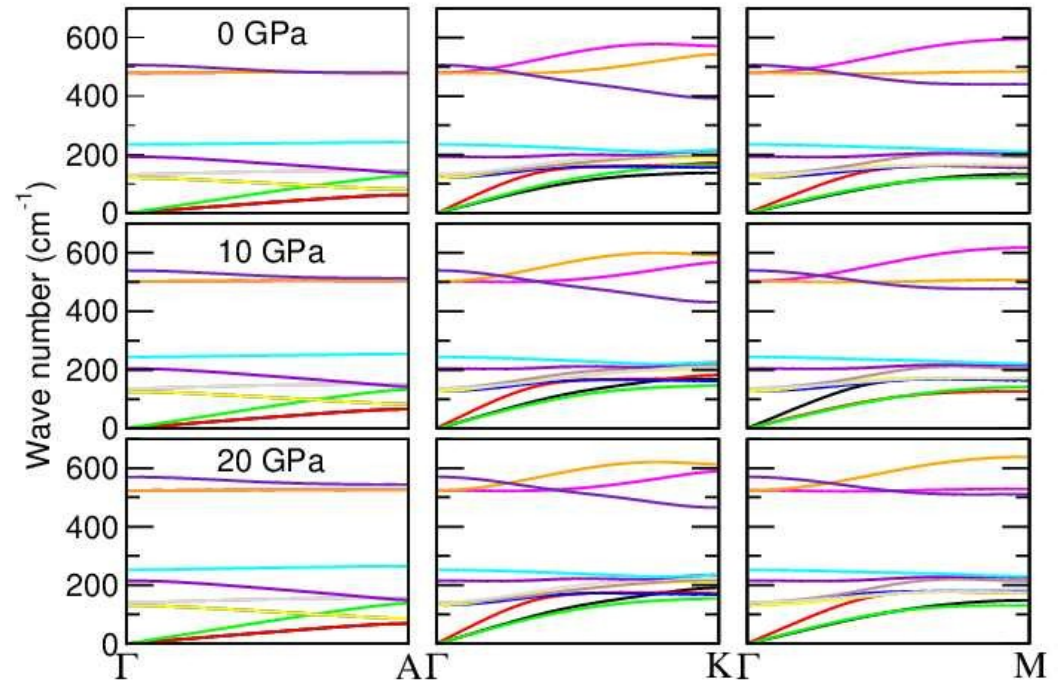
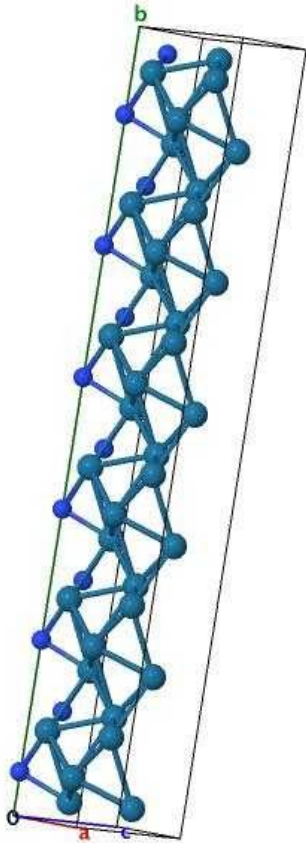


# Approaches to first principles Lattice Dynamics

## Finite displacement and supercells

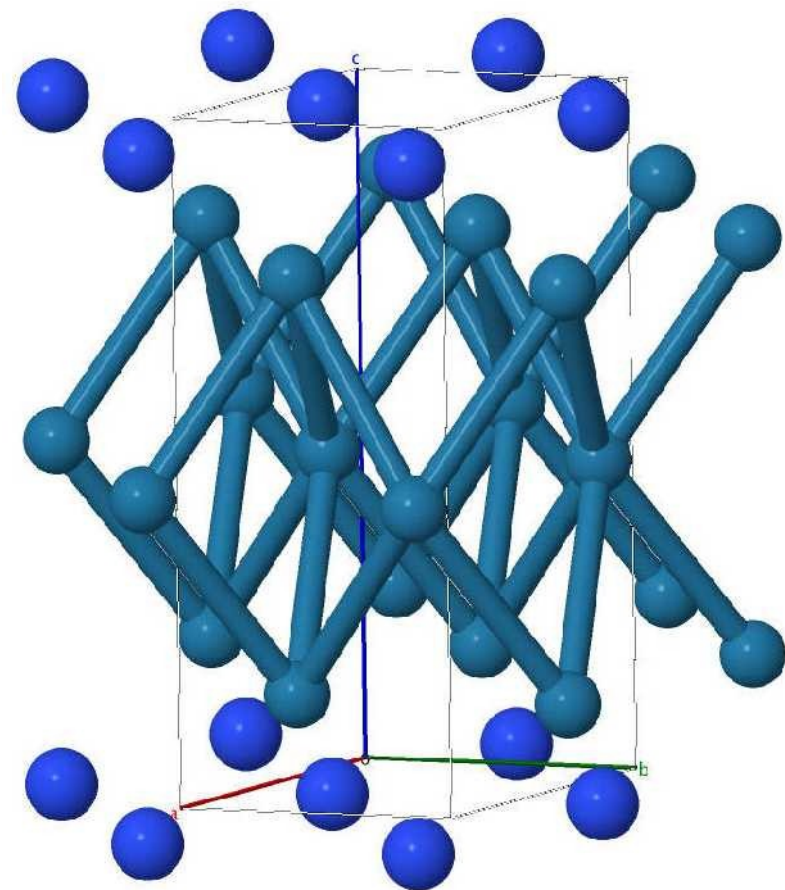
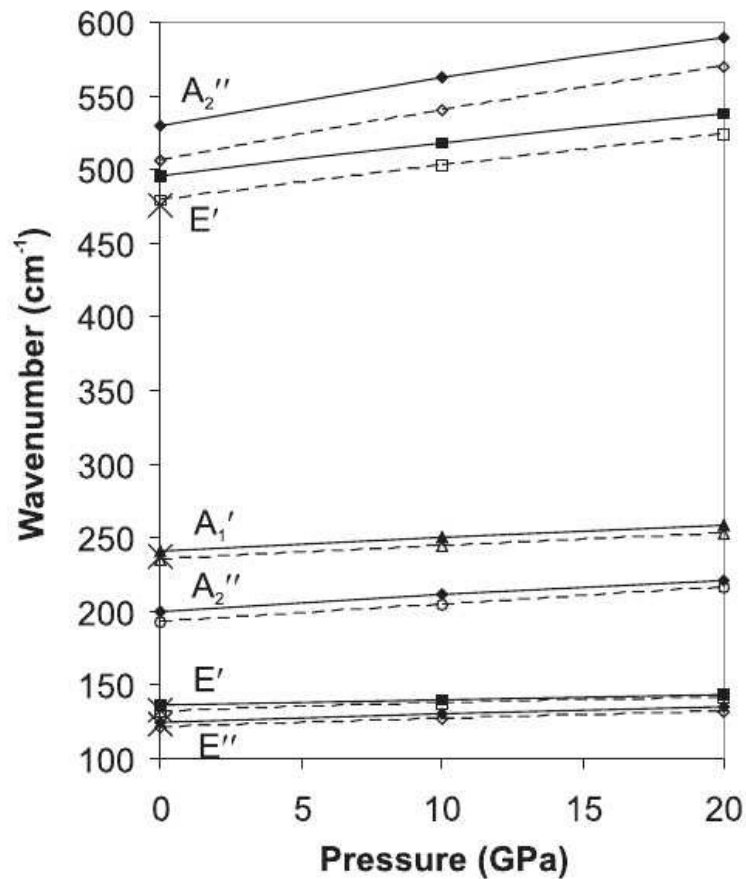
- Works with most general Hamiltonians (PBE0/B3LYP, LDA+U, DFT+D, USPs)
- Accurate, but computationally expensive
- Can not compute dielectric permittivity or Born effective charges
- Can not include LO/TO splitting.
- Can not compute IR or Raman absorptivity/activity

# Re<sub>3</sub>N at high pressure



*Friedrich, et al Phys. Rev. B 82, 224106 (2010).*

# High Pressure - $\text{Re}_3\text{N}$



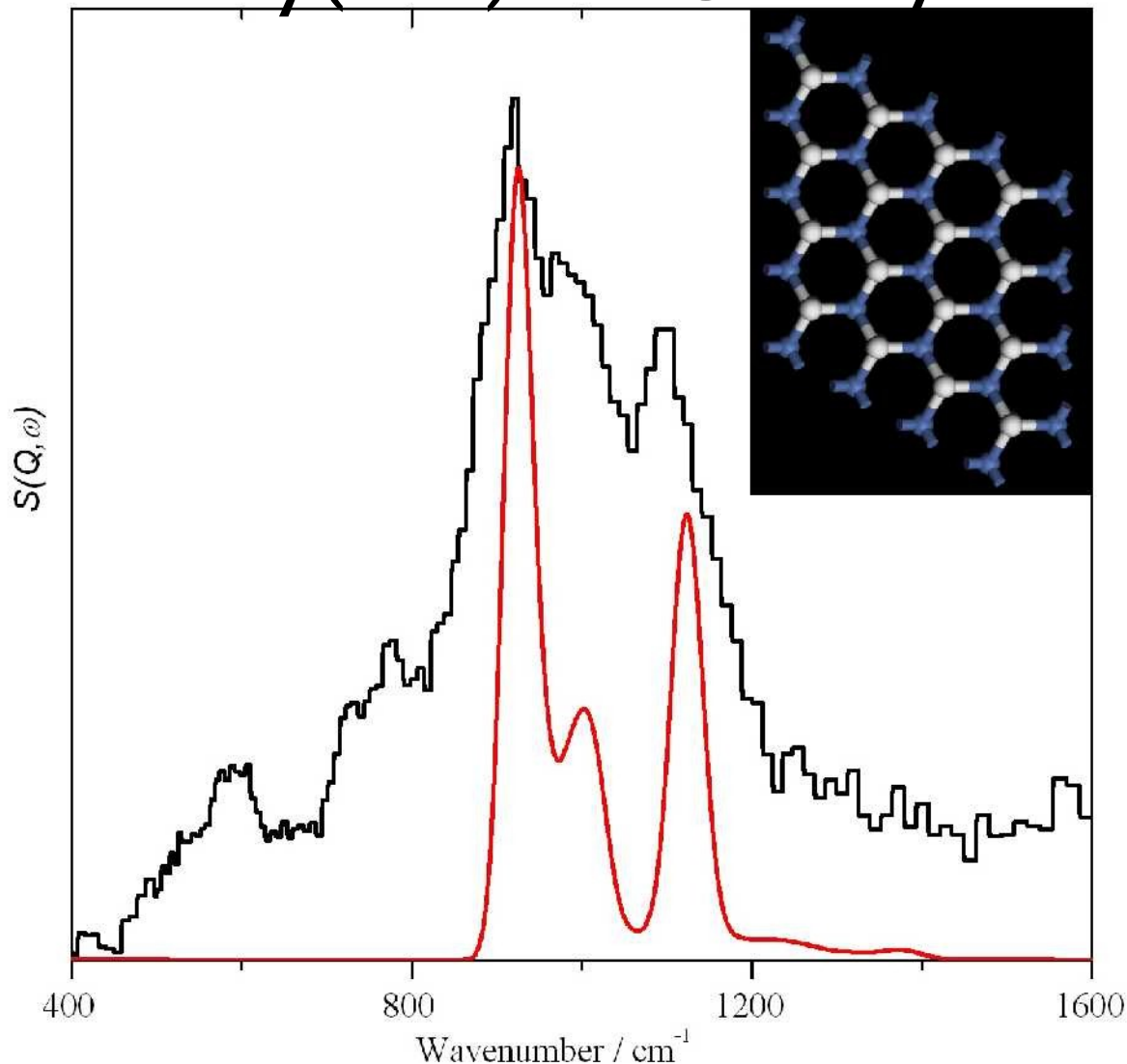
Phys Rev B 82, 244106 (2010)

# Raney(TM) Ni Catalyst

INS spectrum of  
hydrogen on Raney Ni.

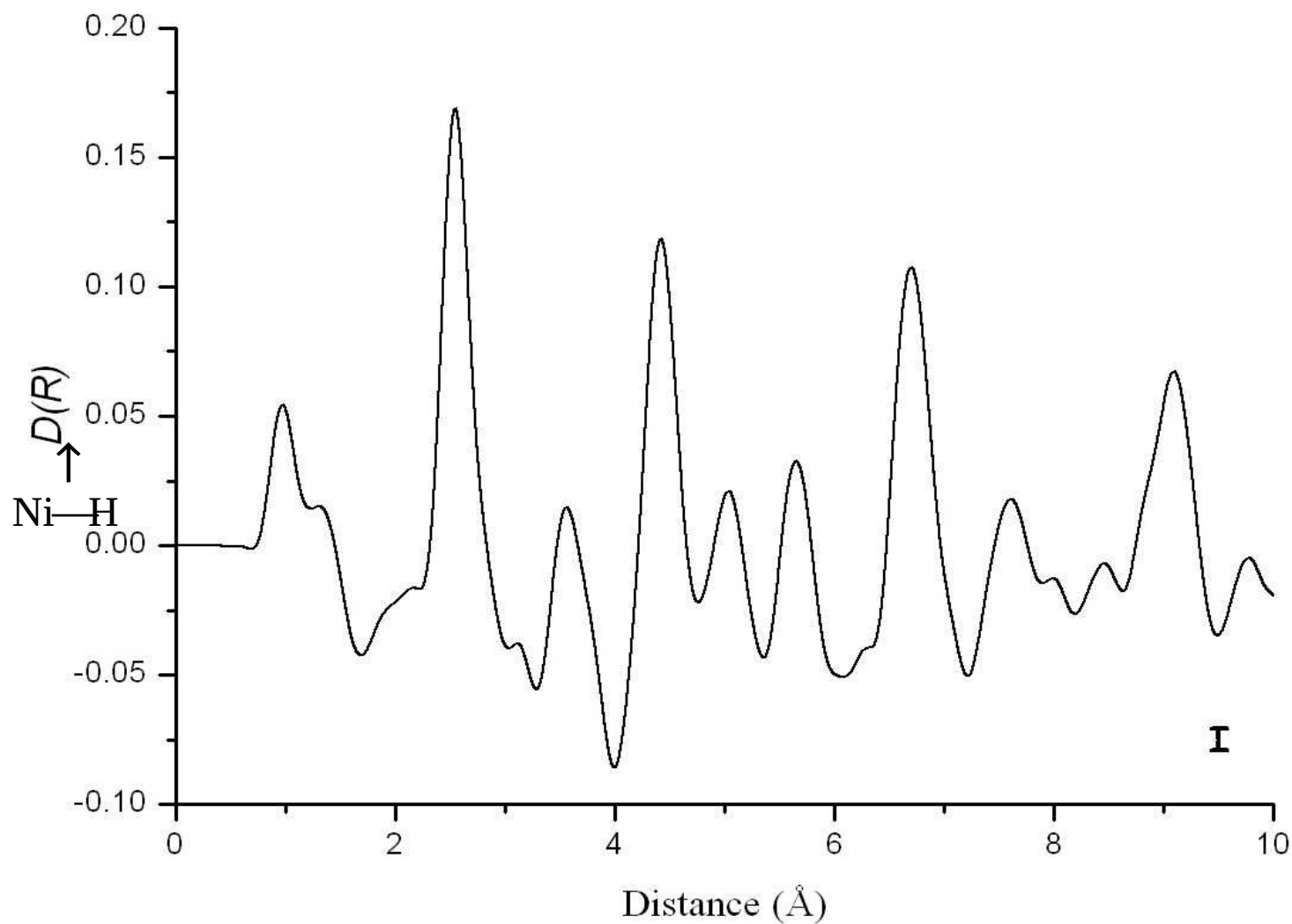
expt (TOSCA)

Full dispersion calc.  
for Ni(111)/H  
(scaled by 0.93).



Stewart Parker, *et al* Chem. Comm. **46**, 2959-61 (2010).

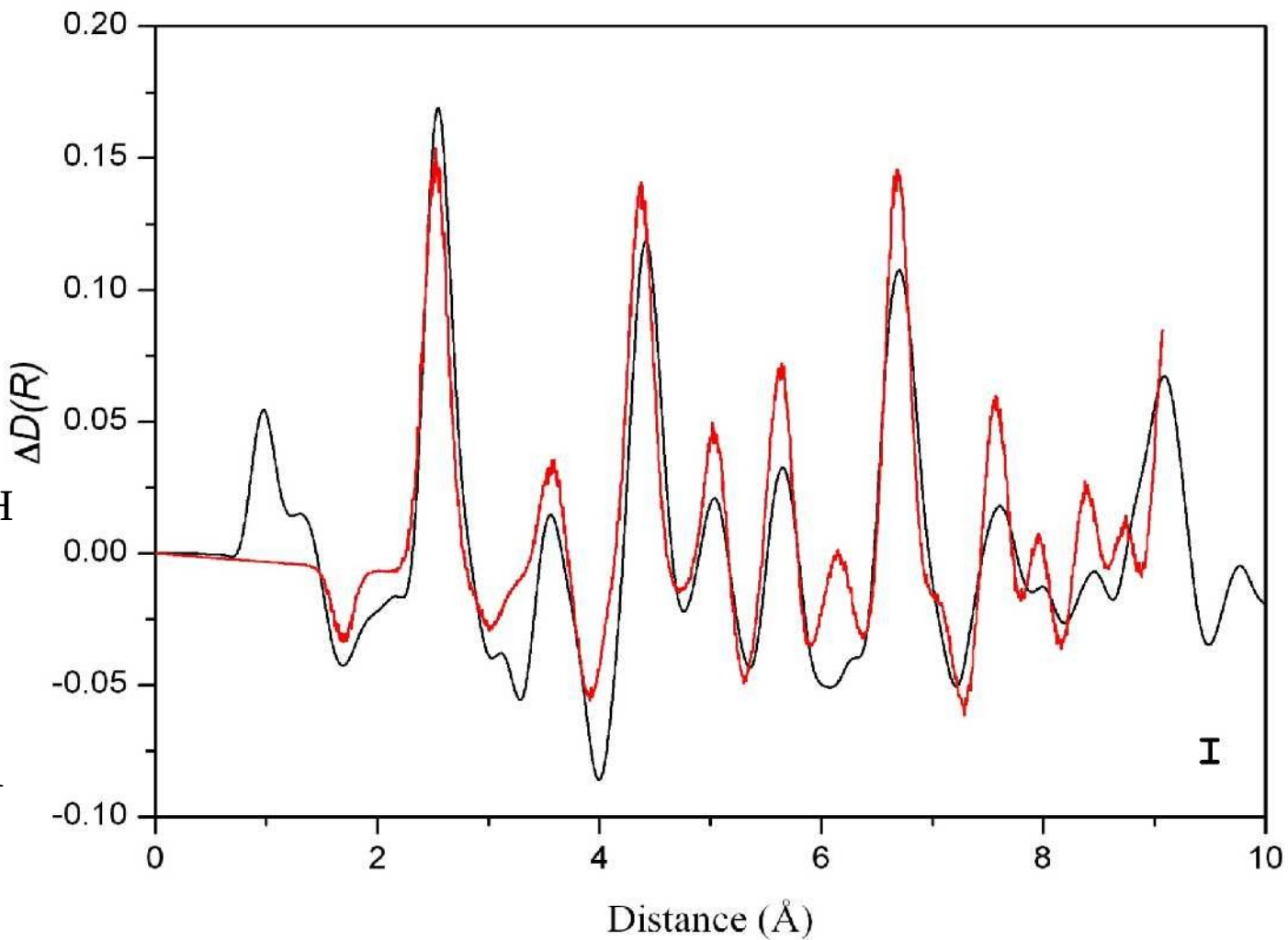
Pair distribution function,  $D(R)$ , for hydrogen on Raney nickel.  
Fourier transform of the normalised difference between the two  $S(Q)$ .





Ni—H  
Neutron: 1.68 Å  
Ab initio: 1.68 Å  
LEIS:  $1.65 \pm 0.05$  Å  
LEED:  $1.84 \pm 0.06$  Å

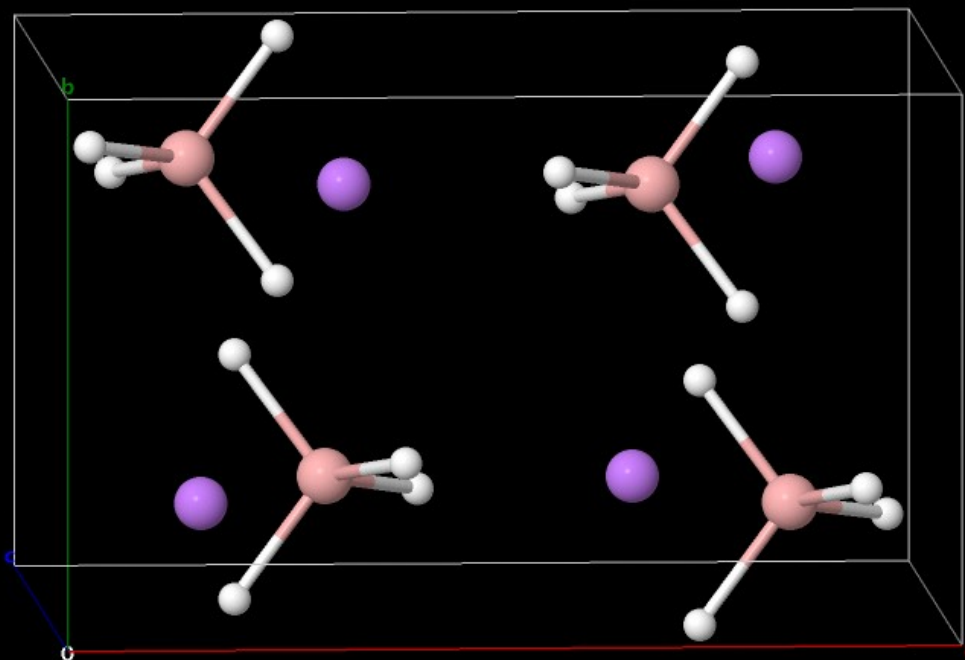
Ni—H  
↓



# H-storage materials

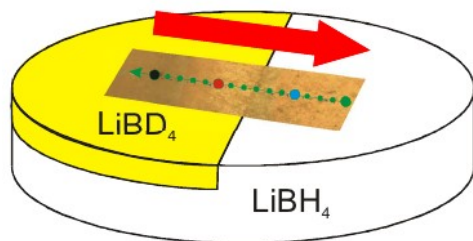
## H transport in LiBH<sub>4</sub>

P 1 [P 1]  
a=7.179Å  
b=4.437Å  
c=6.803Å  
 $\alpha=90.0^\circ$   
 $\beta=90.0^\circ$   
 $\gamma=90.0^\circ$



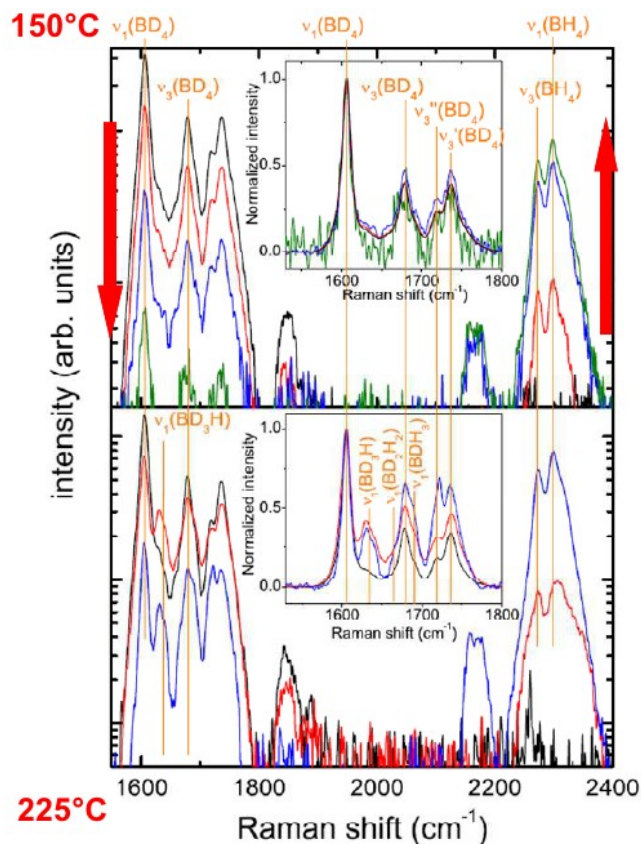
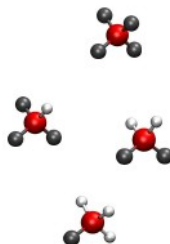
# Raman studies of H transport

Expts: A. Borgschulte, EMPA  
With A.J. Ramirez Cuesta, ISIS



**@ low temperature**  
transport of intact  $\text{BH}_4$  units

**@ >225°C**  
all  
isotopomers  
are present

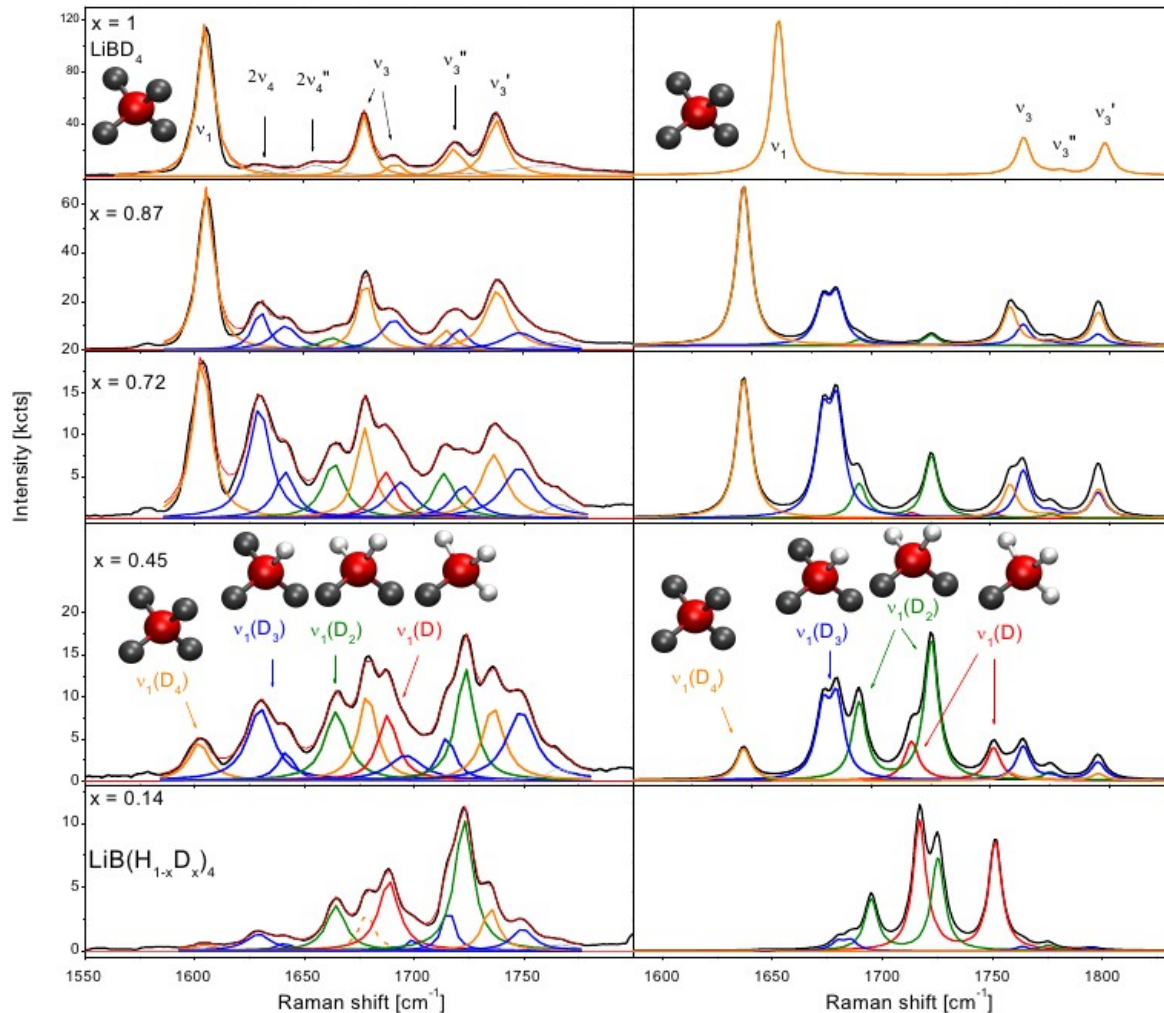




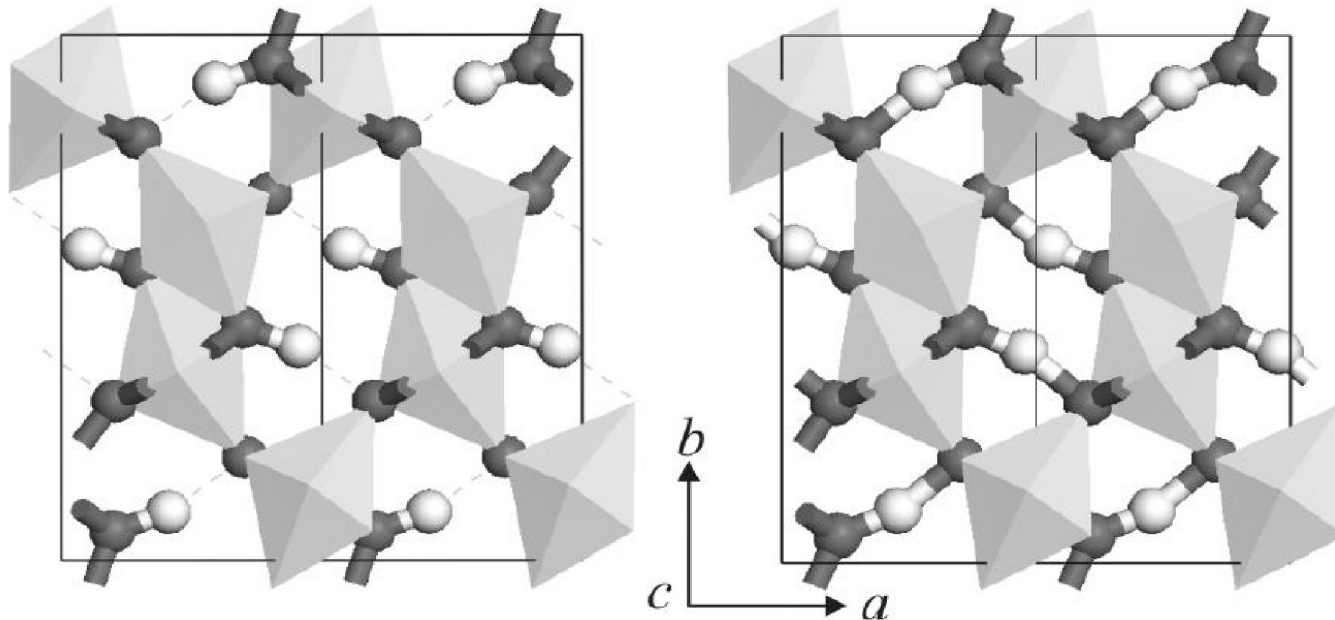
# CASTEP Raman calculation

Measurement

Calculation



# Diaspore (AlOOH)



Canonical example of hydrous mineral

A. Friedrich, D. J. Wilson, E. Hausshl, B. Winkler, W. Morgenroth, K. Refson, and V. Milman, *Phys. Chem. Miner.* 34, 145 (2007).

# Inelastic X-Ray scattering of diaspore

Conventional wisdom: OH groups interact weakly and modes show little dispersion.

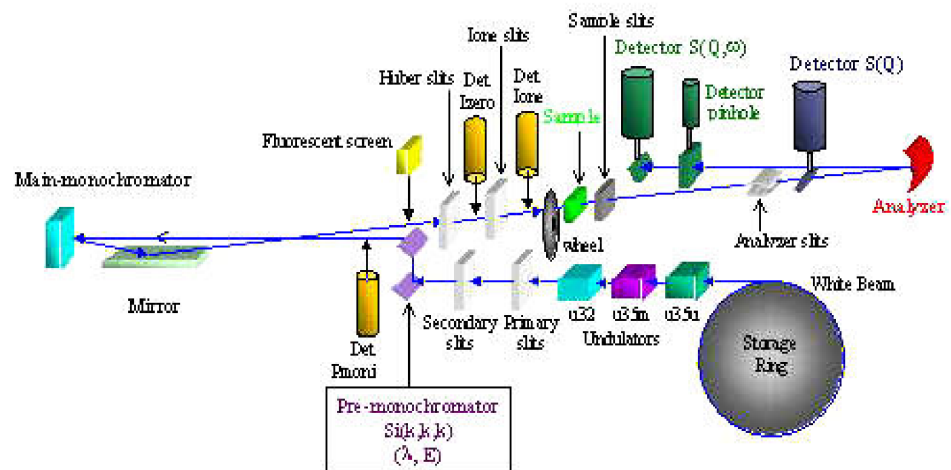
Ab-initio calculations frequently show significantly dispersive OH stretch modes.

INS is not feasible in OH stretch range of 3000–4000  $\text{cm}^{-1}$ .

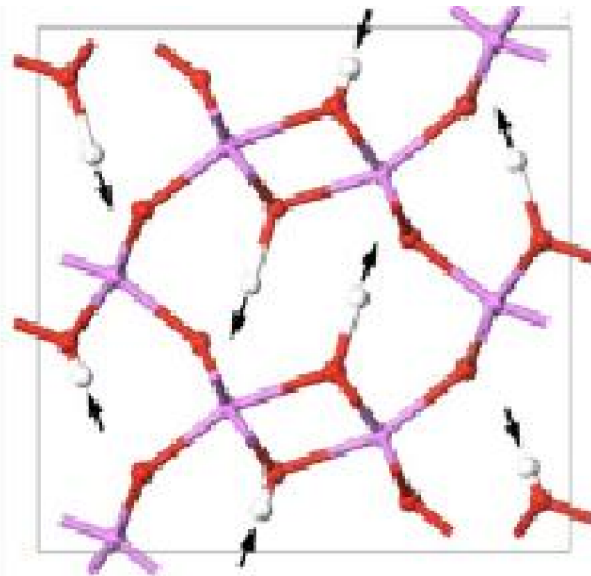
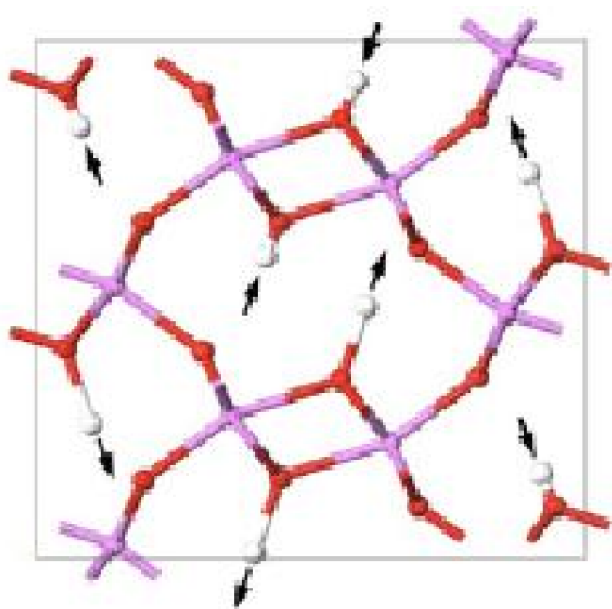
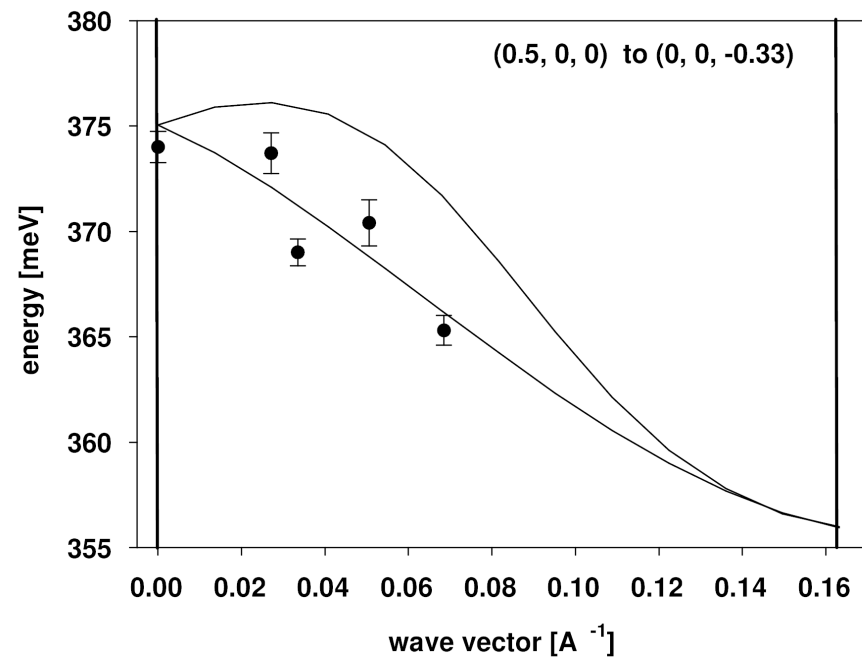
First IXS measurement of OH stretch dispersion ever performed at ID28 at ESRF

B. Winkler, *et. al.*  
 Physical Review Letters,  
 101, 065501 (2008).

## Inelastic X-ray Scattering Beam-line ID28



# OH dispersion in diaspore



## HPC performance and scaling

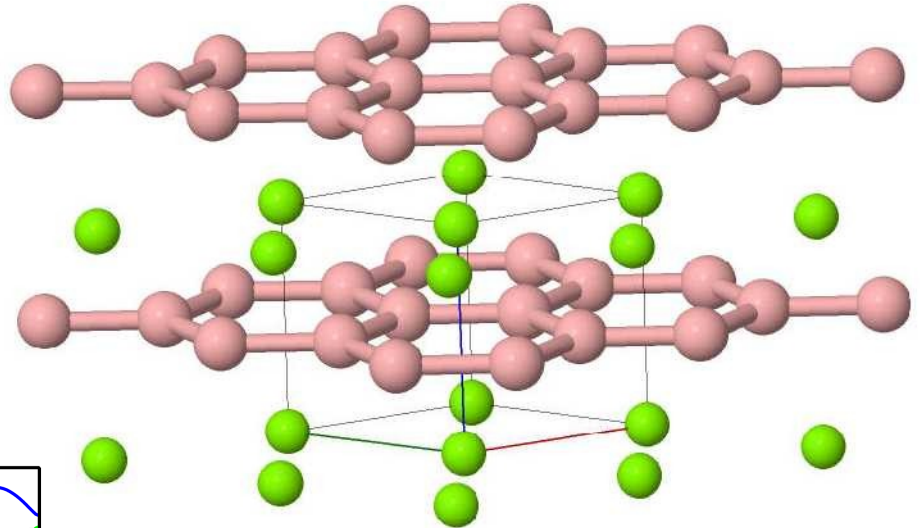
- Band parallelism
- Shared-memory node optimizations of FFT
- Wavefunction read/write optimizations
- Memory optimization
- Distributed  $\langle \beta | \phi \rangle$

## Recently added properties and capabilities

- DFT+D
- Hirshfeld atomic Charges
- NMR Hyperfine Coupling
- New GGAs (PBEsol, WC)
- Hybrid functionals (PBEh, B3LYP), SX
- LDA+U
- EELS and XANES with d core-hole final states
- Raman and infrared intensity/spectroscopy
- Electron localization functions (ELF)
- DFPT phonons for metallic systems

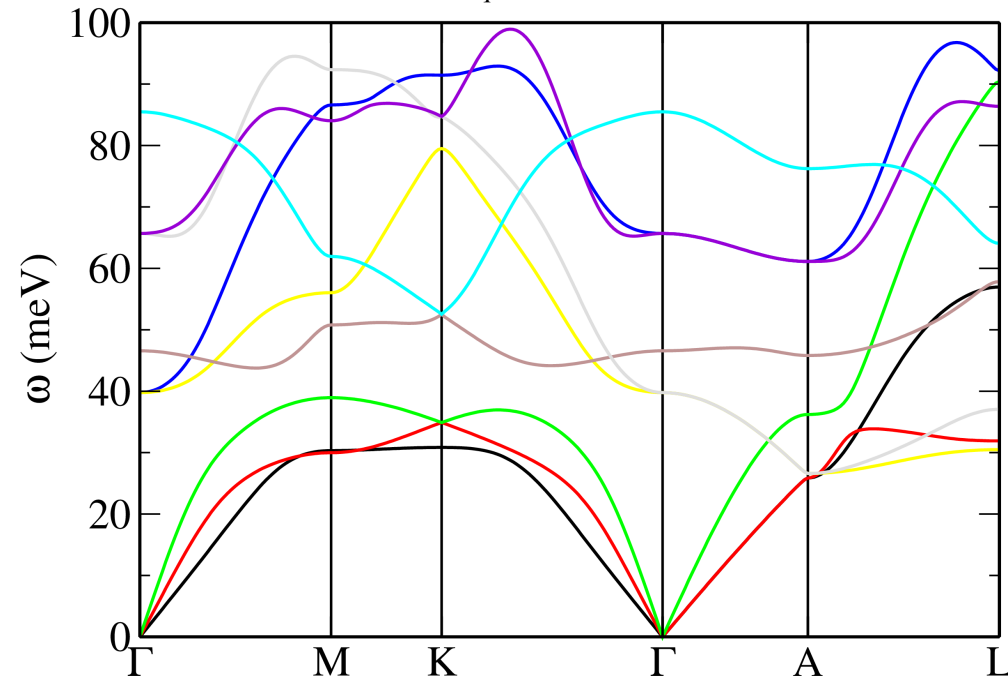
# DFPT Phonons for Metals

P 1 [P 1]  
a=3.085Å  
b=3.085Å  
c=3.519Å  
 $\alpha=90.0^\circ$   
 $\beta=90.0^\circ$   
 $\gamma=120.0^\circ$



MgB<sub>2</sub>

LDA q=5x5x5 k=15x15x12



- Metallic phonon DFPT
- Fourier interpolation for full DOS
- and dispersion calculations.

# Under Development

Much faster Raman activities using DFPT

DFPT for magnetic systems

Non-collinear magnetism

GW for many-body perturbation theory  
Treatment of excitations



# Acknowledgements

Stewart Parker, Timmy Ramirez Cuesta & Mark Adams,  
Daniel Bowron, Alan Soper, Jon Taylor, Steve Bennington  
(ISIS)

Andreas Borgschulte (EMPA)

Boern Winkler, Alexandra Friedrich and Dan Wilson  
(Frankfurt)

HECToR funded from EPSRC (UKCP)

STFC E-Science Facility (SCARF)